Studying The Electrical Properties Of Piezoelectric Material (PZT) Prepared By Organic Acid Precursor Method

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Received on: 5/1/2009  
Accepted on: 16/2/2010

Abstract

The compound of piezoelectric nanopowders with a general system PbZr_{x}Ti_{1-x}O_{3} (PZT) at different concentrations (x =0.3, 0.5 and 0.7) were prepared by using organic acid precursor method. Disc shaped samples of each concentrations were sinteread for two hours at temperatures in the range of (700-1100°C). The perovskite structure was determined by X-ray diffraction spectroscopy, and the dielectric properties were investigated for all samples. The dielectric studies of the compounds as function of temperature in the range (room temperature to 450°C) at frequency 1 KHz shows that the compounds undergo a phase transition of diffuse type from tetragonal phase to cubic phase. Maximum values of electrical properties such as (Dielectric constant 1200, loss factor 1.05) were found for PZ_{0.5} T_{0.5} compounds, and these properties decrease with increasing of (Zr or Ti) additives. The effect of frequency in range (10^{3}-10^{5} Hz) on electrical properties is low with stability in dielectric constant and dielectric loss.

Keywords: PZT ceramics, Dielectric properties, Piezoelectric.

الخلاصة

حضرت مركبات كهربائية اجهادية نانوية ذي الصيغة العامة PbZr_{x}Ti_{1-x}O_{3} (PZT) عند (x=0.3, 0.5, 0.7) بطرق خليط الحامض العضوي. صنعت نماذج بشكل أفرع لمختلف الالترابز حيث أجريت عليها عملية التبلد عند درجات حرارة (700-1100°C). وباستخدام طيف حيود الأشعة السينية تم تحديد تركيب البيروفسكايتس للنماذج المحسورة ومن ثم فحص ودراسة خصائص العزل الكهربائي للعينات كدالة لدرجة الحرارة في المدى (درجة حرارة الغرفة الى 450°C) عند تردد 1 KHz والتي أظهرت بأن المركبات تعاني انقلالات طوريا من الزبيدي إلى المكعب. كما تم الحصول على أعلى قيمة للخصائص الكهربائية مثل (ثابت العزل 1200 و عامل الفقد 1.05) عند النظام (PZ_{0.5} T_{0.5}) عند تردد 10^{4} - 10^{5} Hz. تأثير التردد على الخصائص الكهربائية كان واضحا ضمن المدى من الاضافة (Zr) أو (Ti). Tأثير التردد على الخصائص الكهربائية كان واضحا ضمن المدى من الاضافة (Zr) أو (Ti).
1. Introduction

Piezoelectric materials can generate an electric charge with an applied load of pressure; conversely, they can change their physical dimensions with an application of an electric charge. Conventional ceramic method (involve solid state reaction) and wet-chemical methods are generally utilized in the preparation of piezoelectric ceramic powders. Many materials, both natural and man-made exhibit the piezo-electric effect. These materials include naturally occurring crystals (Quartz, Rochelle salt) and man-made ceramic: the family ceramics with perovskite structures piezoelectric ceramic are in use today. In addition, there are some polymer compounds as piezoelectric [1, 2, and 3].

Lead Zirconate Tetanate (PZT) ceramics is widely used in piezoelectric applications, because of its superior piezoelectric properties. It covers a broad range of applications based on dielectric piezoelectric, pyroelectric, electro-optic and ferroelectric specialty properties, these applications include; firstly, piezoelectric ceramic is used in dielectric industry due to its high resistivity. Secondly: It is used in industry of ceramic electronic elements which are used in electric circuits due to high storage for power inside small volume, [4, 5].

Lead Zirconate Titanate is a binary solid solution of (PbZrO₃) an antiferroelectric (orthorhombic structure) and a ferroelectric (tetragonal perovskite structure). PZT has a perovskit structure with Ti⁴⁺ and Zr⁴⁺ ions, which occupies the B site randomly. The two endmembers are (PT) that undergoes at (490°C) a simple phase transition from a paraelectric cubic to a ferroelectric tetragonal phase and (PZ) that behaves in a more complex fashion having a transition from cubic to an antiferroelectric orthorhombic phase [6,7].

In the present investigation the studies concerned on the electrical properties of a nano particle piezoelectric (PbZrₓTixO₃) compounds with different concentration at x=0.3, 0.5 and 0.7 prepared by organic acid precursor method.

2. Experimental Techniques

Organic acid precursor method is used to preparation of PZT [8], and increasingly utilized to synthesis various ceramic powders. Preparation method of PZT powders can be described as follows; In order to obtain PZT powders, the first step is preparing aqueous solution of (PbCl₂). In the second step, PZT powders are synthesized by adding (ZrO₂ and TiO₂) powders to the aqueous solution of (PbCl₂). Aqueous solution of oxalic acid is added to the result of step 2. The mole ratio of three compounds was 1:1:1.

The result of the above three steps is called "precursor". Then the mixture is stirred and heated up to 80°C till a clear viscous resin is obtained. The precipitates are separated from the mother liquor by a filter-paper for twice with de-ionized water; the washed precipitates were oven-dried at 100 °C for two hours. The powders were pressed at about 2500 kg / cm² for 10 min. into disks 10.1 mm in diameter and 2.4mm in thickness. The samples are calcined in the atmosphere air for two hours at 700 °C. The samples are grind and pressed, then sintered at 900 °C and...
1100 °C for two hours. All samples were polished after sintering. Electrodes on samples surface were coated by silver paste burning method. The poling process has been realized in transformer oil, bath under a DC electrical field of 1.5 kV/mm applied for 10min. at room temperature.

3. Result and Discussion
3.1. XRD Results and Analysis

3.1.1. XRD Pattern
Figure (1) shows the XRD pattern of PZ$_{0.5}$T$_{0.5}$ prepared in this work by chemical method at temperature 700 °C for two hours. The observed XRD peaks correspond to standard PZT patterns with low intensites. Some of peaks refer to PbO, PbCO$_3$ and TiO$_2$ phases. To clear strange phase, high temperature sintering above 700 °C was used.

Figure (2 (a-c)), show the XRD patterns of PZ$_{x}$T$_{1-x}$ sample with various compositions of $x = 0.3$, 0.5 and 0.7 respectively, which are prepared at sintered temperature 1100 °C for 2 hours. From Table (1) data of prepared compounds show a polycrystalline tetragonal PZT structure. The analysis illustrates the reflection from planes (001), (100), (110), (111) and (200) for each concentration. The peak value at (d-spacing = 2.85 Å and 2Θ = 31.36°) of the material PZT represents the major peak (110) which highest degree of crystallinity. Table (1), also shows that the values of (d-space) shifted to large values with the decreasing of Zr content $x$.

3.1.2 Structural Parameters
Structural parameters of unit cell in crystal structure (c, a), are calculated using tetragonal equation for all peaks for (PZ$_x$T$_{1-x}$) at ($x = 0.3$, 0.5 and 0.7).

\[ d_{hkl} = \frac{a}{\sqrt{h^2 + k^2 + l^2}} \]

Table (2) shows values of volume (V) and c/a ratio. The ratio c/a > 1 suggests that PZT has perovskite structure with tetragonal phase.

3.1.3 Grain Size Measurement
The Scherer formula as mentioned below in equation (2) is used to estimate the crystallite size for the prepared samples. Ultra fine powder (in nano size) was obtained from present preparation process of PZT, it is clear from Table (3) that increasing Zr content (x) leads to decrease the grain size of PZT and shows large grain size of PZT at ratio (PZ$_{0.3}$T$_{0.7}$).

\[ t = \frac{0.9 \lambda}{B \cos \theta} \]

3.2 Microstructure
In order to study the microstructure of PZT (grain size and distribution), optical microscope is used. The optical photo-graphs for PZT grains which were sintered at 1100°C for 2hours, are illustrated in Figure (3 (a-c)), for (PZ$_{0.3}$ T$_{0.7}$), (PZ$_{0.5}$ T$_{0.5}$) and (PZ$_{0.7}$ T$_{0.3}$) respectively. These figures show different shapes of grains, some of them are irregular polygons shape, circular shape nearly, spherical and hexagons shape grains. Also the figures below illustrate the broad band distribution of ultra fine grains size.

The crystallite sizes that are determined in this image have an ultra fine microstructure with sub micrometer grains and that indicates
to the absence of abnormal grain growth.

3.3 Bulk Density

In order to study the sinterability of the PbZr$_x$Ti$_{1-x}$O$_3$ system, the bulk density of the sample was measured as a function of sintering temperature with different (x) as illustrated in Figure (4). This figure shows that, the bulk density increases rapidly with temperature and reaches a maximum value at 1100 °C.

This figure also shows that the change in concentration of (x) for (PZ$_x$T$_{1-x}$) system is having small effect on bulk density.

3.4. Dielectric Constant and Temperature

The temperature dependence of the dielectric constant ($\varepsilon_r$) of (PZ$_x$T$_{1-x}$) system for (x = 0.3, 0.5 and 0.7) at 1 kHz is shown in Figure (5). It can be seen that the dielectric constant initially increases gradually with increasing temperature and shows a peak value of around ($\varepsilon_r$ = 1000, 1200 and 1050) at (x = 0.3, 0.5 and 0.7) respectively.

The dielectric constant is reached maximum values at transition temperature from ferroelectric (tetragonal phase) to paraelectric (cubic phase), which is called Curie temperature ($T_c$), for (PZ$_0.3$T$_{0.7}$), (PZ$_0.5$T$_{0.5}$) and (PZ$_0.7$T$_{0.3}$). Table (4) shows values of dielectric constant at Curie temperature for different (x).

With increasing concentration of Zr, the transition temperature shows a shift towards smaller values of temperature as shown in Table (4), which exhibits a characteristic of relaxer material with diffuse phase transition. The broadening in phase transition is attributed to the structural disorder and compositional fluctuation present in the arrangement of cation at A-site and B-site with lattice vacancies, leading to microscopic heterogeneity in the composition and thus results in the distribution of different local Curie points. The ferroelectric transition broadening will be caused by random in domains orientation.

The lower dielectric constant in the present investigation is caused by the effect of the ultra fine nano grain size (30-39 nm).

When the grain size decreases to values comparable to the width of domain walls, pinning points would develop inside the grain and the domain wall motion would be inhibited. The reduced wall mobility will cause a decrease in the relative permittivity.

3.5: Dielectric Loss and Quality Factor with Temperature

Figure (6) shows the variation in dielectric loss of (PZ$_x$T$_{1-x}$) system for (x = 0.3, 0.5 and 0.7) with temperature at frequency of 1 kHz. The dielectric loss exhibits a peak near the transition temperature, the values of dielectric loss for all concentration below Curie temperature, are approximate, while the values increase near the transition temperature region.

The dielectric loss peaks for the three concentrations can be divided into three parts according to their Curie temperature. The first part represents the contribution of orientation mechanism only, where, the contribution of the other mechanisms decreases as a result of decreasing the density due to heating.

The second part is at Curie temperature where the polarization of orientation mechanism is approximately complete, while the third part above Curie temperature where the alignment of orientation dipoles is over due to thermal...
vibration that leads to perturb the alignment. The last part comes from the contribution of the residual polarization mechanisms which decreases with temperature. The former reasons also affect the quality factor (Q). Figure (7) shows that, the quality factor reaches (Q =10, 12.5 and 20) at (x =0.7, 0.5 and 0.3) for (PZT) system.

3.6. Resistivity with Temperature

Figure (8) illustrates resistivity as a function of temperature for (PZT) at (x = 0.3, 0.5 and 0.7). With transition turn to cubic symmetry by increasing temperature, the resistivity transition turn from low amount (below Curie temperature), to high amount (above Curie temperature). That is caused where the dipoles in paraelectric state were over.

3.7 Dielectric Constant with Frequency

Figure (9) illustrates that relative dielectric constant of all the samples as a function of frequency with range of (10^3 - 10^5 Hz) at room temperature. It can be seen that the dielectric constant decreases gradually with increasing frequency. The fall in dielectric constant arises from the fact that the types of polarization don't occur instantaneously with the application of the electric field as charges possess inertia. The delay in response towards the impressed alternating electric field leads to loss and hence decline in dielectric constant.

At frequency of (10^3 - 10^5 Hz) the external field has a large effect on charges and dipoles movement, the three types of polarization mechanisms (orientation, Ionic and electronic) obey the external electric field while orientation polarization at that range contributes to polarization.

3.8. Dielectric Loss and Quality Factor with Frequency

Figure (10) shows the variation of dielectric loss of all the samples with frequency at range of (10^3 - 10^5 Hz) at room temperature. It can be seen that dielectric loss decreases with increasing frequency and exhibits approximate stability in these ranges.

Figure (11) shows the variation of quality factor with frequency, quality factor increases gradually with increase in frequency. Over a wide range of frequencies it is nearly constant.

4. Conclusions

Based on experimental results presented in this search, it is possible to draw the following conclusions. That PZT phase begins forming at temperature 700°C while complete reaction which leads to growth and stability phase occurs at temperature above 700°C. That effect replacement of Zr by Ti leads to falling (The values of 2θ were shifted to large values, these shifts occurred by d-spacing change and full in the grain size for PZT). Maximum values of electrical properties such as (Dielectric constant, loss factor) were found for PZ0.5 T0.5, and these properties decrease with increases Zr or Ti. Bulk density increases rapidly with increase in temperature. The electrical properties depended on nano PZT grains and the high density, the broad dielectric constant and dielectric loss peaks which indicate the domain orientation is random and this tends to broaden the ferroelectric transition, this is an important property in microwave devices applications. That effect of frequency
in range (10^3-10^5 Hz), on electrical properties is low and stability in dielectric constant and dielectric loss, where this stability is an important property in tunable devices applications.

5. References:

[8] Abdel Aal, MM Rashad and GA Amin "Dielectric Thin Film from Barium Titanate Nanpowders"
**Table (1): Comparison ($2\theta^\circ$ and $d\ A^\circ$) between JCPDS and PZ$_x$T$_{1-x}$ for ($x = 0.3$, 0.5 and 0.7) prepared at 1100C°**

<table>
<thead>
<tr>
<th>STD.For,PZ$<em>{0.5}$T$</em>{0.5}$</th>
<th>X=0.3</th>
<th>X=0.5</th>
<th>X=0.7</th>
</tr>
</thead>
<tbody>
<tr>
<td>hkl</td>
<td>$2\theta^\circ$</td>
<td>$d\ A^\circ$</td>
<td>$2\theta^\circ$</td>
</tr>
<tr>
<td>100</td>
<td>22</td>
<td>4.03</td>
<td>22.66</td>
</tr>
<tr>
<td>110</td>
<td>31.36</td>
<td>2.85</td>
<td>31.43</td>
</tr>
<tr>
<td>111</td>
<td>38.25</td>
<td>2.35</td>
<td>39.07</td>
</tr>
<tr>
<td>200</td>
<td>44.87</td>
<td>2.01</td>
<td>43.73</td>
</tr>
</tbody>
</table>

**Table (2): Values of (c/a) and ($V\ A^3$) for PZ$_x$T$_{1-x}$**

<table>
<thead>
<tr>
<th>(PZ$<em>x$T$</em>{1-x}$)</th>
<th>c/a</th>
<th>V ($\ A^3$)</th>
</tr>
</thead>
<tbody>
<tr>
<td>X=0.3</td>
<td>1.032</td>
<td>65.92</td>
</tr>
<tr>
<td>X=0.5</td>
<td>1.029</td>
<td>66.41</td>
</tr>
<tr>
<td>X=0.7</td>
<td>1.029</td>
<td>66.43</td>
</tr>
</tbody>
</table>
Table (3): the grain size measurement for the prepared samples.

<table>
<thead>
<tr>
<th>PZ₃₅T₁₋ₓ</th>
<th>c. s. (nm)</th>
</tr>
</thead>
<tbody>
<tr>
<td>X=0.3</td>
<td>39</td>
</tr>
<tr>
<td>X=0.5</td>
<td>36</td>
</tr>
<tr>
<td>X=0.7</td>
<td>30</td>
</tr>
</tbody>
</table>

Table (4) Values of Dielectric constant at Curie temperature for PZₓ T₁₋ₓ

<table>
<thead>
<tr>
<th>PZₓ T₁₋ₓ</th>
<th>Dielectric constant (εᵣ)</th>
<th>Curie temperature (°C)</th>
</tr>
</thead>
<tbody>
<tr>
<td>X=0.3</td>
<td>1000</td>
<td>440</td>
</tr>
<tr>
<td>X=0.5</td>
<td>1200</td>
<td>370</td>
</tr>
<tr>
<td>X=0.7</td>
<td>1050</td>
<td>320</td>
</tr>
</tbody>
</table>

Figure (1) XRD patterns of PZ₃₅T₁₋₅ prepared at 700°C for 2h.

O=PbCO₃, V=PbO, ⊙=TiO₂
Studying The Electrical Properties Of Piezoelectric Material (PZT) Prepared By Organic Acid Precursor Method

![Graph](image1)

**Figure (2-a) XRD patterns of PZ\(_x\)T\(_{1-x}\) prepared at 1100 °C for 2h. for (a) X=0.3**

![Graph](image2)

**Figure (2-b) XRD patterns of PZ\(_x\)T\(_{1-x}\) prepared at 1100 °C for 2h. for (b) X=0.5**
Figure (2-c) XRD patterns of PZ$_x$T$_{1-x}$ prepared at 1100 °C for 2h. for (c) X=0.7
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![Figure (3): Optical microscope image for PZ$_x$T$_{1-x}$, M= 600X for (a) X=0.3, (b) X=0.5 an (c) X=0.7](image1)

Figure (4) Variation bulk density with temperature for (PZ$_x$T$_{1-x}$) at x=0.3, 0.5, 0.7

![Figure (5): Variation of Dielectric Constant $\varepsilon_r$ with Temperature $^\circ$C at 1 KHz for (PZ$_x$T$_{1-x}$) at x= 0.3, 0.5 and 0.7](image2)

Figure (6) Variation of Dielectric Loss ($\tan\delta$) with Temperature ($^\circ$C) at 1 KHz for (PZ$_x$T$_{1-x}$) at x=0.3, 0.5 and 0.7
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Figure (7) Variation of Quality Factor with Temperature at 1 KHz for \( (PZ_x T_{1-x}) \) at \( x=0.3, 0.5 \) and 0.7

Figure (8) Variation of resistivity \((\Omega \cdot \text{cm})\) with temperature \(^\circ\text{C}\) at 1 KHz for \( (PZ_x T_{1-x}) \) at \( x=0.3, 0.5 \) and 0.7

Figure (9) Variation of Dielectric Loss \((\tan \delta)\) with Frequency (Hz) for \( (PZ_x T_{1-x}) \) at \( x=0.3, 0.5 \) and 0.7

Figure (10) Variation of Dielectric Loss \((\tan \delta)\) with Frequency (Hz) for \( (PZ_x T_{1-x}) \) at \( x=0.3, 0.5 \) and 0.7

Figure (11) Variation of Quality Factor with Frequency (Hz) for \( (PZ_x T_{1-x}) \) at \( x=0.3, 0.5 \) and 0.7